

Novel Graphene Oxide Based Adsorbent Embedded with Epichlorohydrin and Allyl Amine Hydrochloride (GO-ECH-AAH) were Prepared and Employed for Endotoxin Removal from Aqueous Solutions

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ABSTRACT

A facile strategy to synthesis GO-ECH-DETA nanocomposite by the reaction of graphite oxide (GO) with epichlorohydrin (ECH) as a coupling agent and allylamine hydrochloride (AAH) as a cationic ligand (GO-ECH-AAH), were prepared and employed for endotoxin accumulation and removal from aqueous solution. In this work, the synthesis and characterization of GO-ECH-DETA nanocomposites through seeded emulsion polymerization and the selective light reflection properties of dry films have been reported. Amphiphilic molecule sulfonated 3-pentadecyl phenol was used as a co-surfactant to stabilize GO dispersions during the emulsion polymerization process. The synthesis mechanism of the nanocomposite was studied and characterized with Scanning Electron Microscopy (SEM), X-ray diffraction patterns, Transmission Electron Microscopy (TEM), Fourier Transform Infrared Spectroscopy (FTIR), X-ray Photoelectron Spectroscopy (XPS), Raman spectroscopy, UV-Vis spectra and Atomic Force Microscopy (AFM) were used to study the morphology and structure of the composite particles on drying. The AFM study confirms the non-spherical shape of the particles. This kind of nanocomposite undoubtedly shows promising applications in the fabrication of multi-functional materials and can be used as a potential candidate for removal of PLS from aqueous solution. Batch adsorption studies showed that the product possesses superior adsorption capacity of endotoxin from aqueous solution. This is a simple and cheap procedure, has proven to remove endotoxins without affecting any significant losses in protein yields and biological activities. The physicochemical properties of the nanocomposite were fully characterized, adsorption equilibrium and kinetic analysis indicated that the adsorption isotherm was well fitted by Langmuir isothermal model with the maximum adsorption capacity of 138.63 mg g⁻¹, the kinetics of the endotoxin adsorption process was shown to follow the pseudo-second-order model. The results showed that the optimum condition for endotoxin removal was obtained at pH of 6.05, GO-ECH-AAH dosage of 500 mg L⁻¹, contact time of 60 min and endotoxin concentration of 200.0 endotoxin units per milliliter (EU mL⁻¹).

Keywords: Graphene oxide, Epichlorohydrin, Allylamine hydrochloride, Nanocomposites, Endotoxin, Adsorption

INTRODUCTION

Lipopolysaccharides (LPS), also called endotoxins, are one of the main pollutants found in commercially produced proteins and biologically active substances, which often interfere with biological effects of the main ingredient. The presence of small amounts of endotoxin in recombinant protein preparations causes tissue injury, endotoxin shock and even death. Thus, it is often essential to remove endotoxins from drugs, injectable and other biological and pharmaceutical products. LPS are derived from cell membrane of Gram-negative bacteria and are responsible for its organization and stability. In pharmaceutical industries endotoxins always found in the final product. Endotoxins release does not happen only with cell death but also liberated during growth and division into the surrounding environment. Even if the media is poor in nutrients, such as

clean or saline water, endotoxins are found almost everywhere. However, upon cell death LPS are shed in large amount than during growth and division. They are highly heat-stable and are not destroyed under regular sterilizing conditions. However, LPS can be inactivated when exposed

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at temperature of 250°C for more than 30 min or 180°C for more than 3 h [1,2]. In laboratory scale, acids or alkalis concentration of at least 0.1 M strength fair enough to destroy endotoxin [3]. A single *Escherichia coli* estimated to contain about 2 million LPS molecules per cell. Endotoxin causes a variety of wide pathophysiological effects. If small concentrations of endotoxin enter the human blood stream, it can cause a systemic inflammatory reaction, leading to multiple pathophysiological effects, such as endotoxin shock, tissue injury, and even death [4-6]. It is important to note that, LPS act through activation of the immune system, especially through monocytes and macrophages, with the release of a range of pro-inflammatory mediators, such as tumor necrosis factor (TNF), interleukin (IL)-6 and IL-1. Pyrogenic reactions and shock are seen in mammals upon intravenous injection of endotoxin at low concentrations (1 ng/mL) [7]. The maximum level of endotoxin for intravenous applications of pharmaceutical and biologic product is set to 5 endotoxin units (EU) per kg of body weight per hour by all pharmacopoeias [8]. The term EU describes the biological activity of an endotoxin. For example, 100 pg of the standard endotoxin EC-5 and 120 pg of endotoxin from *Escherichia coli* O111:B4 have activity of one EU [9]. Meeting this threshold level has always been a challenge in biological research and pharmaceutical industry [7,8]. It should be mentioned that endotoxins may also have beneficial effects. They have been used in artificial fever therapy, to destroy tumors and to improve, non-specifically,

the immune defense. The uncertainty about its role for the human health was once described by Bennett [9]. On the other hand, any superfluous endotoxin exposure, especially for intravenously-administered medicines must be strictly avoided to prevent complications.

In the pharmaceutical industries and biotechnology, recombinant DNA products such as peptides and proteins are produced by the Gram-negative bacteria such as *Escherichia coli*. These products are usually contaminated with endotoxins [10]. For this reason, these proteins must be purified from endotoxin, to prevent any side effects induced when administered to humans. However, LPS are very stable molecules, resisting to extreme temperatures and pH values in comparison to proteins [11-13]. Many different procedures have been used for the accumulation and removal of LPS from proteins. These include two-phase extractions, affinity resins, ultrafiltration, membrane adsorbers, hydrophobic interaction chromatography and ion exchange chromatography. These techniques provide different efficiency degrees of LPS separation from proteins, is highly dependent on the properties of the protein of interest. LPS structure composed of a hydrophilic polysaccharide moiety, which is covalently linked to a hydrophobic lipid moiety (Lipid A) (**Figure 1**) [10]. Endotoxins from most species are composed of three distinct regions: the O-antigen region, a core oligosaccharide and Lipid A (LipA) as depicted in **Figure 1**.

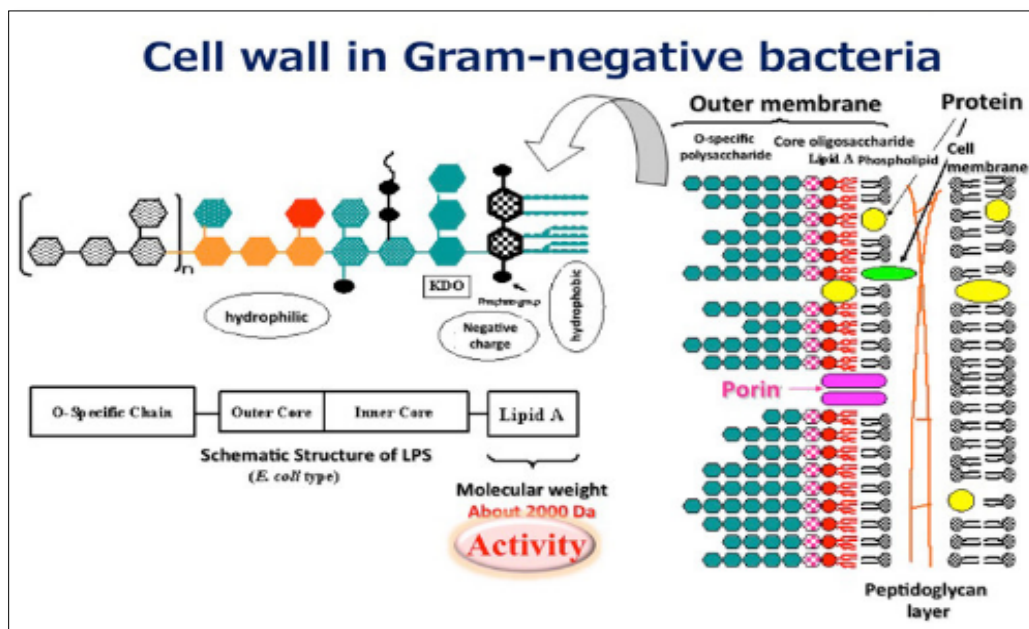


Figure 1. Chemical structure of LPS from *E. coli* O111:B4.

The most biological activate part of endotoxin is lipid A and indeed is responsible for its toxicity. Endotoxin is composed of b-1,6-linked D-glucosamine residues, covalently linked to 3-hydroxy-acyl substituents with 12-16 carbon atoms via amide and ester bonds. These further esterified with

saturated fatty acids. This hydrophobic part of endotoxin adopts an ordered hexagonal arrangement, resulting in a more rigid structure compared to the rest of the molecule [12,14]. The core oligosaccharide has an inner 3-deoxy-D-manno-2-octulosonic acid (KDO) - heptose structure region

and an outer hexose region. In *E. coli* species, five different core types are known and *Salmonella* species share only one core structure. The core region close to lipid A and lipid A itself are partially phosphorylated ($pK_1=1.3$, $pK_2=8.2$ of phosphate groups at lipid A). Thus endotoxin molecules exhibit a net negative charge in common protein solutions [10]. The O-antigen is generally composed of a sequence of identical oligosaccharides (with three to eight monosaccharides each), which are strain specific and determinative for the serological identity of the respective bacterium [8]. The endotoxin monomer molar mass range from 10 to 20 kDa, owing to the variability of the oligosaccharide chain; even extreme masses of 2.5 (O-antigen-deficient) and 70 (very long O-antigen) kDa can be found. There are various forms of endotoxins supra-molecular aggregates in aqueous solutions due to their amphipathic structures. Based on molecular dynamics, the three-dimensional structure of endotoxin, especially the long surface antigen, is much more flexible than the globular structure of proteins [13]. These aggregates result from non-polar interactions between lipid chains as well as of bridges generated among phosphate groups by divalent cations [1]. The aggregate structures have been studied by different techniques such as electron microscopy, X-ray diffraction, FT-IR spectroscopy and NMR. Results from these studies have shown that, in aqueous solutions, endotoxins can self assemble in a variety of shapes, such as lamella, cubic and hexagonal inverted arrangements, with diameters up to 0.1 μ m and 1000 kDa and high stability depending on the solution characteristics (pH, ions, surfactants, etc.) [1,2].

The old approved techniques used by FDA for endotoxin detection are the rabbit pyrogen test and *Limulus* Amoebocyte Lysate (LAL) assay [15,16]. The rabbit pyrogen test, is an old technique used in 1920s, involves measuring the rise in temperature of rabbits after intravenous injection of a test solution. Due to its high cost and long turnaround time, the use of the rabbit pyrogen test has diminished and is now only applied in combination with the LAL test to analyze biological compounds in the earlier development phase of parenteral devices. Today the most popular endotoxin detection systems are based on LAL, which is derived from the blood of horseshoe crab, *Limulus polyphemus* and clots upon exposure to endotoxin. The simplest form of LAL assay is the LAL gel-clot assay. When LAL assay is combined with a dilution of the sample containing endotoxin, a gel will be formed proportionally to the endotoxin sensitivity of the given assay. The endotoxin concentration is approximated by continuing to use an assay of less sensitivity until a negative reaction (no observable clot) is obtained. This procedure can require several hours [16]. The concentration of 0.5 EU/mL was defined as the threshold between pyrogenic and non-pyrogenic samples [16].

In addition to the gel-clot technique, scientists have also developed two other techniques: turbidimetric LAL

technique and the chromogenic LAL technique. These newer techniques are kinetic based, which means they can provide the concentration of endotoxin by extracting the real-time responses of the LAL assay. Turbidimetric LAL assay contains enough coagulogen to form turbidity when cleaved by the clotting enzyme, but not enough to form a clot [17]. The LAL turbidimetric assay, when compared to the LAL gel-clot assay, gives a more quantitative measurement of endotoxin over a range of concentrations (0.01 EU/mL to 100.0 EU/mL). This assay is based on the turbidity increase due to protein coagulation related to endotoxin concentration in the sample. The optical densities of various test-sample dilutions are measured and correlated to endotoxin concentration helped by a standard curve obtained from samples with known amounts of endotoxin [18] kinetic chromogenic substrate assay differs from gel-clot and turbidimetric reactions because the coagulogen is partially or completely replaced by a chromogenic substrate [19]. When hydrolyzed by the pre-clotting enzyme, the chromogenic substrate releases a yellow-colored substance known as *p*-nitroaniline. The time required to attain the yellow substance is related to the endotoxin concentration [18]. However, kinetic turbidimetric and chromogenic tests, although more accurate and faster than the gel-clot, cannot be used for fluids with inherent turbidity such as blood and yellow-tinted liquids, e.g. urine and their performance may be compromised by any precipitation from solution [19]. Therefore, different new methods for detection of endotoxin in different samples have been studied and approved [20,21].

Carbonaceous nanomaterials, including carbon nanotubes (CNTs) and graphene based materials (GBMs) such as graphene oxide (GO), hold significant promise in engineering and medicine due to their intrinsic electro-mechanical properties. Graphene and graphene oxide are the most basic form of carbon, it is composed of sp^2 bonded carbon atoms arranged in a hexagonal arrangement in a 2D plane [22]. The lattice of graphene consists of two interleaved triangular shaped carbon sub lattices. The sub lattices overlap in such a way that carbon atom from one sub lattice is at the centroid of the other sub lattice. Graphene has been utilized in many engineering and industrial applications and graphene-based polymer nanocomposites exhibit superior promising properties. For example, graphene-based polymer composites show better thermal, mechanical and electrical properties than the normal polymer [23,24]. It has been shown that the mechanical and electrical properties of graphene-based polymer composites are much better in comparison to clay or other carbon filler-based polymer composites [25-29]. One of the main applications of graphene sheets is use as reinforcement agents for the preparation of nanocomposites with different types of polymers. Other than mechanical properties, electrical and thermal properties of the polymeric matrix can also be enhanced. It is a fact that the graphene-based nanocomposites present improved properties compared to

the original raw form of graphene. Graphene nanocomposites with polysaccharides such as chitosan have many diverse new applications. Polysaccharide exist both as linear or branched polymers, since their repeating monosaccharide units are connected via O-glycosidic bonds [30]. Their properties, including gelation, water solubility and other surface properties depend on the type of monosaccharide composition. Advantages such as abundance in nature, biocompatibility, biodegradability, easy functionalization and relatively easy isolation from their natural sources have led to their study and use in several applications, especially in the field of drug delivery and biomaterials [31]. Another application of graphene derivatives with polysaccharide is the use in accumulation and removal of various types of pollutants from wastewater effluents [32-34]. Graphene nanocomposites with chitosan have been used for the removal of dyes [35,36], heavy metal ions [32-34] and pharmaceutical compounds [32-34] from aqueous solutions. Despite the intriguing properties of polysaccharides, their poor mechanical properties limit their applications. Nanofillers such as graphene are known to improve the properties of raw polymers, not only the mechanical but also the thermal and electrical properties [37,38]. Moreover, the effects of the incorporation of graphene and graphene oxide together in raw polymers have been extensively studied, mostly synthetic polymers reinforced by graphene and graphene oxide find several improvement in properties such as mechanical strength, thermal stability, gas barrier properties, electrical and thermal conductivity, etc. [39-46].

Cationic polymers which are useful as flocculants are prepared by condensation reaction of a dialkylamine, dicyandiamide and a polyalkylenepolyamine, with a difunctional epoxide. Functionalization of graphene oxide (GO) by crown ether moiety to attach Li^+ in the cage of five oxygen is a useful tool to achieve 2D material for Li ion battery. The attachment of crown ether occurs only through the reaction with epoxy groups of GO. Epichlorohydrin is used to increase the number of the epoxy group to enhance and precise control of the Li^+ content for tuning the activation energy of Li^+ migration [47].

The final properties of nanocomposites depend on various factors; the most important is the interfacial bonding between the filler and the matrix. Poor adhesion can lead to aggregates of the nanofillers or gaps between the surface of the composites components, acting as stress concentration points and therefore causing premature failure of the materials. Besides, the compatibility between nanofiller and matrix, the geometrical and the aspect ratio of the fillers play a similarly important role. Graphene possesses a high surface area, high aspect ratio and high strength which are reasons for the enhanced performance of its nanocomposites. Large graphene or GO flakes with high surface areas have proved to be more efficient reinforcing agents than similar structures with smaller aspect ratio.

Some commonly used techniques have been extensively used for removing endotoxin contaminants are ultrafiltration [48] and ion exchange chromatography. Ultrafiltration has been successfully effective in removing endotoxins from water. Nevertheless, universal adoption of this technology is limited by the presence of proteins, which can be damaged by physical forces [49]. Anion exchangers, which take advantage of the negative net charge of endotoxins, have been used for endotoxin adsorption. However, when negatively charged proteins need to be decontaminated, they may co-adsorb onto the matrix and cause a significant loss of biological material. Also, net-positively charged proteins form complexes with endotoxins, causing the proteins to drag endotoxin along the column and consequently minimizing the endotoxin removal efficiency [50]. LPS removal is more efficient on cationic exchangers than on anionic exchangers. In recent years, alkane diols were shown to be effective agents for the separation of LPS from LPS-protein complexes during chromatography with ionic supports. Their effectiveness in reducing the protein complexation with LPS is dependent on (I) the size of the alkanediol, (II) the isomeric form of the alkanediol, (III) the length of the alkanediol wash, (IV) the concentration of alkanediol and (V) the type of ionic support used, cationic or anionic. Membrane-based chromatography has been successfully used for LPS separations from protein, universal adoption of this technology has not taken place because membrane chromatography is limited by the binding capacity, which is small when compared to that of bead-based columns, even though the high flux advantages provided by membrane adsorbers would lead to higher productivity [51-60].

Jann et al. [61] tested that slab-polyacrylamide gel electrophoresis in the presence of sodium dodecyl sulfate (SDS-PAGE) can be used for the separation of LPS. Several methods have been used to separate the different subclasses of LPS from individual strains, with sodium dodecyl sulfate-polyacrylamide gel electrophoresis (SDS-PAGE) and gel filtration being perhaps the most successful Tsai and Frasch [62], McIntire et al. [63], Oroszlan and Mora [64], Weiser and Rothfield [65], Ribi et al. [66], Hannecart-Pokorni et al. [67], McIntire et al. [68], Kim et al. [69], Morrison and Leive [51], Maccari and Volpi [70] and Dietrich and Dietrich [71]. These methods are hampered by the tendency of LPS to aggregate and by the difficulty in detecting and identifying each distinct subclass [72]. Reichelt et al. [73] reported that the removal of endotoxin could be achieved during chromatography purification with the use of Triton X-114 in the washing steps. The application of 0.1% Triton X-114 in the washing steps was successful at reducing endotoxins during histidine and GST (resin GST sepharose) fusion protein purification, whereas washing steps lacking surfactant were ineffective in eliminating endotoxins. In contrast to purified materials employing the standard protocol which contained from 2500 to 34000 EU mg^{-1} ,

purified recombinant proteins treated with Triton X-114 contained concentrations as low as 0.2 to 4 EU mg⁻¹ (less than 1% of initial endotoxin content). Reichelt et al. [73] studied whether the use of Triton X-114 in washing steps could eliminate endotoxins from proteins with a pI above 8.5. They found that washing with Triton X-114 coupled with affinity chromatography effectively removed endotoxins from negatively-charged proteins (SyCRP and NdhR). The minimal endotoxin concentration achieved was lower than 0.2 EU mg⁻¹; protein recovery and yield were close to 100% [73]. The use of two-phase aqueous micellar systems for the purification or concentration of biological molecules, such as proteins and viruses has been growing [74-76]. In these systems an aqueous surfactant solution, under the appropriate solution conditions, spontaneously separates into two predominantly aqueous, yet immiscible, liquid phases, one of which has a greater concentration of micelles than the other [77]. The difference between the physicochemical environments in the micelle-rich phase and in the micelle-poor phase forms the basis of an effective separation and makes two-phase aqueous micellar systems a convenient and potentially useful method for the separation, purification, and concentration of biomaterials [78]. Particularly for endotoxin removal, above the critical micelle concentration (CMC) of surfactants, endotoxins are accommodated in the micellar structure by non-polar interactions of alkyl chains of lipid A and the surfactant tail groups and are consequently separated from the water phase (micelle-poor phase). Surfactants of the Triton series show a miscibility gap in aqueous solutions. Above a critical temperature, the so-called cloud point, micelles aggregate to droplets with very low water content, by that forming a new phase. Endotoxins remain in the surfactant-rich phase. Through centrifugation or further increase in temperature the two-phases separate with the surfactant-rich phase being the bottom phase [79,80]. If necessary, this process is repeated until the remaining endotoxin concentration is below the threshold limit. The cloud point of Triton X-114 is at 22°C, which is advantageous when purifying proteins [81] used Triton X-114, showed that a 100-fold endotoxin reduction in two steps with a final endotoxin content of 30 EU mg⁻¹ and 50% loss in bioactivity of the exopolysaccharide. In addition, about 100-fold endotoxin reduction was shown by Cotten et al. [82] from plasmid DNA preparation with a final endotoxin content of 0.1 EU in 6 µg DNA. The detergents, even though they were also very effective at reducing the LPS levels, are relatively expensive, would add significant cost to a manufacturing process, and may affect the bioactivity of the protein of interest. Alternative chemicals are desired that could safely and cost effectively be used in place of the alcohols or detergents as washing agents for the separation of LPS from proteins during chromatographic unit operations [83]. Indeed, these chemicals would be relatively inexpensive, chemically well-defined, present minimal safety issues and ideally have minimal impact on

the bioactivity of the protein in question when implemented into a process.

The objective of this research is to test the synthesis and applications of a new nanocomposite polymer formed by mixing and cross-linking networks graphite oxide (GO) with epichlorohydrin (ECH) as a coupling agent and allylamine hydrochloride (AAH) as a cationic ligand to form (GO-ECH-AAH). Then, we will examine the mechanical properties of the nanocomposite polymer as well as its various applications in the accumulation of PLS. The fabricated GO nanocomposites polymers have better mechanical and thermal properties and stabilities than GO nanocomposites alone. The results has shown strong interactions between the functional groups of the three components, confirmed by FTIR spectra, led to a series of improved properties, including mechanical strength in both wet and dry conditions. The results of this research has shown that the novel graphene oxide (GO)-based adsorbent embedded with epichlorohydrin (ECH) as a coupling agent and allylamine hydrochloride (AAH) as a cationic ligand (GO-ECH-AAH), has excellent endotoxin removal from aqueous solutions and biotechnological preparations. The results of our studies showed that endotoxin strongly loads on the GO-ECH-AAH nanocomposite derivative via σ - σ , π - π and n - π interaction and bonding between the nanocomposite and the LPS. The interaction and adsorption behavior of the prepared composite was elucidated with a series of experiments. The results revealed that the adsorption mechanism dominated between endotoxin molecules and the GO-ECH-AAH matrix favor acidic conditions were the optimum for the adsorption process at pH 5.5. The Langmuir-Hinshelwood kinetic model adequately describes the experimental results; both the pseudo-first order kinetic constants of the reactions and the adsorption constants were calculated. GO-ECH-AAH was more active than GO alone for the endotoxin accumulation. The reduction of 90% of the endotoxin was observed after 1hr. Thus, graphene polymer nanocomposites (GO-ECH-AAH) offer a green alternative to synthetic polymers in the preparation of soft nanomaterials, results indicated that a significant interaction of the allylamine hydrochloride of the AAH with both GO and ECH and ECH-AAH polymer were inserted between the GO layers and (ii) ECH reacted with carboxyl and epoxy groups of GO, leading to its reduction and hence the destruction of the layered structure.

MATERIALS AND METHODS

Go synthesis

GO was synthesized through a modified Hummer's method [84]: 6 g of natural graphite powder (Graphene Laboratories Inc.), 4.5 g sodium nitrate and 207 mL sulphuric acid were added in a reaction flask, kept at 10°C and stirred for 30 min, followed by the addition of 27 g potassium permanganate. The solution was stirred for 45 min and then 414 mL of water was added. After 12 h, 1260

mL of warm water and 45 mL oxygen peroxide (30%) were added. The suspension was filtered, washed several times and finally dried at 60°C in a vacuum oven.

Nanocomposite fabrication

The epoxy resin used in this study was epichlorohydrin (ECH) as a coupling agent and Allyl Amine Hydrochloride (AAH) a ligand and the hardener, both supplied by Huntsman. Nanocomposite samples were prepared without nanomaterials (i.e., neat epoxy resin), with GO and with AAH. They all received the same amount of filling material, 0.25% wt. and the same amount of hardener, 27% wt. To improve GO dispersion, these fillers were each mixed with acetone through bath sonication (25 kHz) for 30 min. An aqueous solution of epichlorohydrin (ECH) (5 wt%) was prepared at 100°C upon stirring for 1 h and subsequently cooled to room temperature. An aqueous solution of Allyl Amine Hydrochloride (AAH) (5 wt%) was prepared at 90°C upon stirring for 2 h and subsequently cooled to room temperature. The Allyl Amine Hydrochloride (AAH) a ligand hardener was added to each mixture and heated to 90°C to melt down. The chemically reduced GO dispersion was then mixed with epichlorohydrin (ECH) epoxy resin solution at 65°C and ultrasonicated (42 kHz) for another additional 1 h. The resulting mixtures were degasified at 80°C for 24 h to eliminate volatiles such as acetone and avoid bubbles in the final nanocomposites. Curing process was held in two stages: at 80°C for 1 h and at 120°C for 2 h, according to the manufacturer's instructions.

Endotoxin detection methods

We used Sun et al. [85] applied cysteamine-modified gold nanoparticles to detect LPS by UV-Vis spectrum and the detection limit was decreased to 3.3×10^{-10} mol/L. More facilely, a colorimetric biosensor fabricated with gold nanorods was developed and can detect LPS in the concentration range of 0.01 to 0.6 μ M. Nanorods of a high aspect ratio were also demonstrated to show superiority in sensing. The facile assay for the rapid visual detection of lipopolysaccharide (LPS) molecules down to the low nanomolar level by taking advantage of the electrostatic interaction between GO and Epichlorohydrin (ECH) and Allyl Amine Hydrochloride (AAH). The large amount of negatively charged groups on the LPS molecules make LPS highly negatively charged. Thus, when modified with cysteamine, the positively charged gold nanoparticles can aggregate in the presence of trace amounts of LPS. The probe is simple, does not require any advanced instrumentation, and the limit of detection (LOD) was determined to be as low as 3.3×10^{-10} mol/L. To the best of our knowledge, it is the most sensitive synthetic LPS sensor reported so far.

Characterization techniques

Raman spectrum was used to show the graphitic ordering before and after functionalization treatments on GO and GO-

ECH-AAH samples. It was acquired on a Renishaw 2000 Micro-Raman, with Ar laser ($\lambda=514.5$ nm) and range of 500-3500 cm^{-1} (only first order spectrum is shown in results). XPS high-resolution spectra were obtained to determine atomic composition of GO in a UNI-SPECS UHV spectrometer (5×10^{-7} Pa, $h\nu=1253.6$ eV). FT-IR was used to characterize the presence of chemical groups on GO surfaces. Infrared spectra were recorded on a Perkin-Elmer Spectrum GX, in the range of 4000-400 cm^{-1} with 4 cm^{-1} resolution, 12 scans and KBr pellet method. X-ray photoelectron spectroscopy (XPS) was employed for the analysis of the surface chemistry of GO and GO-ECH-DETA, using a SPECS system equipped with a Phoibos 150 1D-DLD analyser (Berlin, Germany) and monochromatic Al K α X-ray source (1486.6 eV). The XPS survey-scan spectra were recorded with pass energy of 80 eV, step energy 1 eV and dwell time 0.1 s; whereas the individual high-resolution spectra were collected with pass energy of 30 eV, step energy 0.1 eV and dwell time 0.1 s, at an electron take-off angle of 90°. A Renishaw Invia microscope (Gloucestershire, UK) with laser frequency of 514 nm was used to obtain the Raman spectra of the graphenic materials from 500 to 3500 cm^{-1} . The information about the methods for the structural, morphological, microstructural and thermal characterization of GO and GS is displayed in the **Supplementary Material**. The XRD patterns of graphenic materials, GO and GO-ECH-AAH nanocomposites were performed on a Malvern Panalytical (Almelo, Netherlands) X'PERT PRO automatic diffractometer operating at 40 kV and 40 mA, in theta-theta configuration, secondary monochromator with Cu-K α radiation ($\lambda=0.154$ nm) and a PIXcel solid state detector (active length in 2θ 3.347°). Data were collected in the range of $2\theta=1$ -50° (step size of 0.026° and time per step of 80 s, total time 20 min) at room temperature. A variable divergence slit giving a constant 5 mm area of sample illumination was used. The Bragg equation ($\lambda=2d \sin\theta$) was used to determine the interlayer distance in the graphenic materials. A Hitachi S-4800 scanning electron microscope (Tokyo, Japan) operating at an accelerating voltage of 15 kV was used to obtain SEM images of the neat GO-ECH-DETA nanocomposite films, after being freeze fractured by liquid nitrogen and sputtered with gold. TEM micrographs of nanocomposites were obtained with a Philips Tecnai G2 20 TWIN TEM (Eindhoven, Netherlands) at 200 kV accelerated voltage after cutting the GO-ECH-DETA films into thin sections with a Leica EM UC6 ultramicrotome apparatus, at room temperature and placing the sliced specimens in copper grids. Differential scanning calorimetry analyses were performed by a Mettler Toledo DSC 3+ unit (Greifensee, Switzerland). The samples were heated from -30°C to 250°C at a heating rate of 10°C/min under a nitrogen gas flow of 20 mL/min. Values were obtained from the first cooling and second heating scans. Thermogravimetric analysis was performed on a TA instruments TG-Q-500 (New Castle, DE, USA) at a heating rate of 10°C/min from

40°C to 800°C in nitrogen or air-flow. An electromechanical testing machine (Instron 5967, Norwood, MA, USA) operating at room temperature with a load cell of 500 N, a gauge length of 10 mm and a cross head speed of 5 mm/min was used to performed tensile tests. Films were cut into a dog-bone shape before testing and kept at a relative humidity of 58% at room temperature for more than one week to ensure equilibration of the moisture uptake in the films. Testing was carried out on at least ten identical composite films of each composition and the average values were reported.

RESULTS AND DISCUSSION

X-ray diffraction (XRD) analysis

Figure 2 shows the XRD patterns of GO, ECH-AAH and GO-ECH-AAH nanocomposite. The basal reflection peak (0 0 1) of pure GO at $2\theta \approx 10.00$ is shifted to $2\theta \approx 20.00$ in the GO-ECH-AAH spectrum because of the intercalation of ECH and AAH chains in the interlayer spacing of GO. Both pure ECH-AAH as a cationic ligand and the GO-ECH-AAH nanocomposites exhibit an intense peak at a 2θ value of 22.00, corresponding to a basal spacing of 3.95 Å. This indicates the formation of several layers of GO. The peak at $2\theta \approx 22.00$ is attributable to the formation of graphene oxide structures. The broad peak centered at $2\theta \approx 22.50$ is attributable to the intercalation of ECH epoxy chains between the stacked GO layers. The spectrum of GO-ECH-

AAH nanocomposite shows two amorphous peaks at $2\theta \approx 20.00$ and 22.00 . The absence of characteristic peak of GO in the composites indicates the delamination of GO layers in the presence of ECH epoxy. The XRD data also imply the homogeneous dispersion of GO in the nanocomposites. However, XRD is not the best tool to determine crystal layer delamination or the homogeneity of dispersion. High magnification electron microscope can be used to confirm the homogeneity of the composites. The results from XRD analysis of the sample film formed has a bandwidth on the lower diffraction angle the results from XRD analysis of the sample film formed has a bandwidth on the lower diffraction angle ($2\theta = 1.96^\circ$) corresponding to the existence of oxygen-rich groups on both sides of the leaves and the water molecule inserted between the GO sheets. The existing peaks in the diffractogram similar suggest that the ECH-AAH chains are interspersed between the GO layers. Thus, maintaining the arrangement of the graphene nanosheets, suggesting that after the functionalization of graphene oxide with the ECH and AAH the occurrence of flare spacing d is attributed to covalently bonded interconnection with the GO. In fact, the ECH-AAH is bonded on the surface of GO; can be attributed to the diffraction band orientation of GO sheets with the ECH and AAH chains cross-link's to form a GO-ECH-AAH nanocomposite lattice.

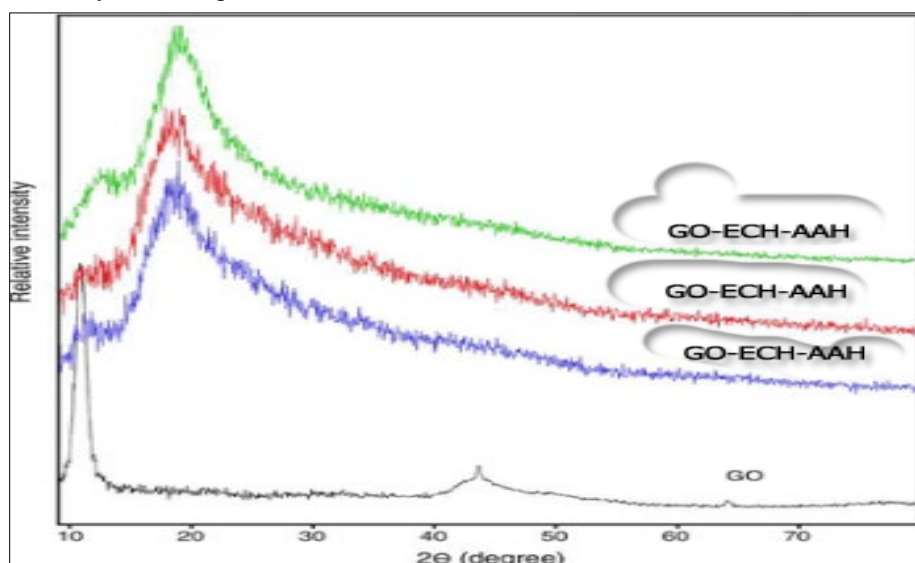


Figure 2. XRD patterns of (a) GO, and b, c and d GO-ECH-AAH nanocomposite prepared at three different temperature.

SEM images

Figure 3 shows SEM images of GO before and after functionalization treatments of GO with ECH-AAH to form GO-ECH-AAH nanocomposite. The polymerized nanocomposite showed a slight degree of tangling, but the blocks were clearly undone and the basic structure of the flakes was preserved. SEM images of GO and GO-ECH-AAH nanocomposite evidence the

good dispersion state of graphene sheets throughout ECH-DETA. Single dispersed sheets and aggregated nanosheets with thickness ~ 12 nm coexist. However, a better degree of dispersion is achieved in the polymerized nanocomposites. From SEM patterns it can be inferred an exfoliated morphology for the samples prepared by the in situ method. For the nanocomposite sample obtained by the *ex situ*

method, a good dispersion and exfoliation of graphene sheets is observed.

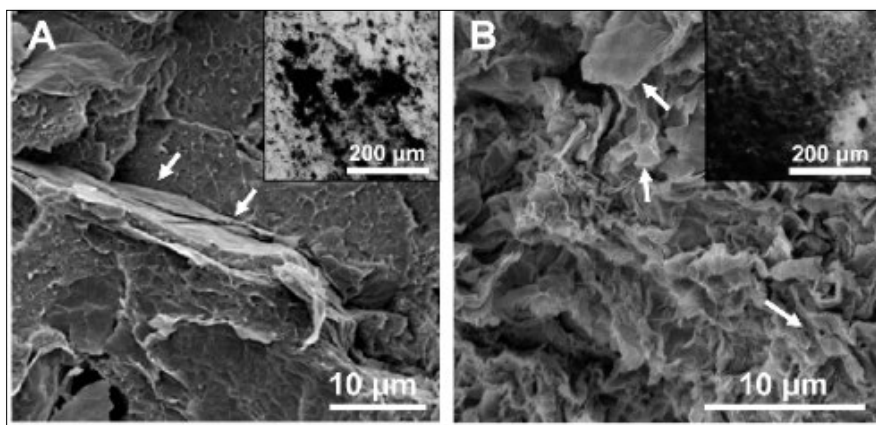


Figure 3. SEM images of GO: (A) a GO before functionalization treatments; (B) after functionalization treatments of GO with ECH-DETA to form GO-ECH-AAH nanocomposite treatment.

TEM images

Figure 4 shows the TEM images of GO before functionalization treatments; (b) after functionalization treatments of GO with ECH-DETA to form GO-ECH-AAH nanocomposite treatment. It indicates a brittle wrinkled fracture of GO. After formulation with ECH-AAH, by solution blending, the GO could be homogeneously dispersed in the ECH-AAH matrix, becomes cloud-like and rough. The cracks become more randomly dispersed, indicating that the GO network acts as an obstacle to crack propagation. In this image the TEM of **Figure 3b** have a display surface in some areas of irregular morphology of fibers and various forms of holes evenly along the fiber, are structures are similar to the rod. But, there are 2 wt.% graphene oxide inclusions longer the polymer, the fibers presented in the fracture morphology is connected; indicating a high physical interaction existing between ECH-AAH and GO nanocomposite. Overall, graphene has carboxylic acid functional group provides an intermolecular force calling itself bridge effect. Providing a cohesive bond on the GO with ECH-AAH system provides a more efficient load

transfer to the polymer matrix. Thus, ECH-AAH addition enhances the strength of the composite. However, further addition of ECH-AAH (beyond 1.5 vol%) leads to the formation of clusters within the GO network that are in the scale of microns. This inhibits the stress transfer from the ECH-DETA matrix to the GO network, thus deteriorating the strength of the composite. It is clear from the SEM image that GO is generally dispersed properly in the matrix. Upon impact, the crack propagates in the direction of the tension, and then proceeds to the weak interfaces, finally damaging the material. At 4.5 vol% ECH-AAH, the fracture surface is non-uniform. The ECH-AAH flow is hindered by the GO agglomerates. The composite surface shows signs of brittleness, with a rougher surface than neat ECH-AAH. The pull-out of the GO in the ECH-AAH matrix is also seen to decrease.

Extremely thin graphene nanosheets with a wrinkled structure were detected. After formation of the nanocomposite with PLA, by solution blending, the graphene nanosheets could be homogeneously dispersed in the PLA matrix.

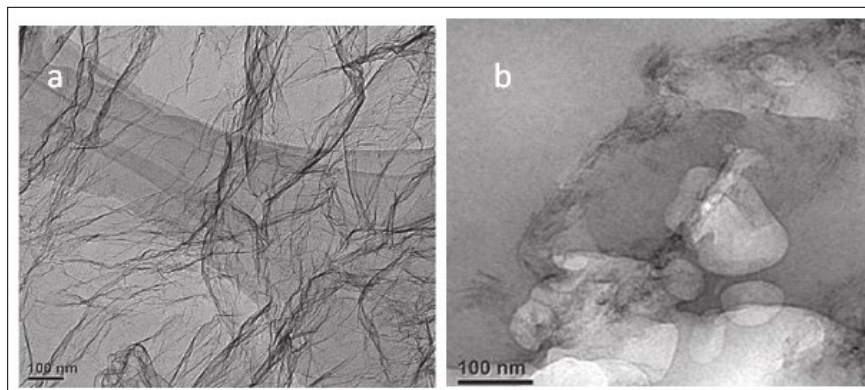


Figure 4. TEM images of GO: (a) a GO before functionalization treatments; (b) after functionalization treatments of GO with ECH-AAH to form GO-ECH-AAH nanocomposite treatment.

Polarized optical microscopy (POM) images

Composites crystallization behaviors via POM are shown in **Figures 5 and 6**. The POM photographs of pure GO and GO-ECH-AAH nanocomposite are shown in **Figure 5**. The thin nanosheet, wrinkled and fine morphology of pure GO is seen in **Figure 5a**. Compared to AAH polymerization crystallization, the GO-ECH-AAH nanocomposite shows faultiness in crystallization after the reaction process was complete. The phenomenon may be due to the intrinsic slow rate of crystallization of GO-ECH-AAH. **Figure 6** shows polarized optical microscopy (POM) images of (a) pure poly

allyl amine hydrochloride (AAL) and (b) GO-ECH-AAH at 65°C during non-isothermal crystallization from their melts at a cooling rate of 2°C/min. Nevertheless, the results suggest that GO can be an efficiency nucleation accelerator for AAH, it is heterogeneous nucleating agent during the non-isothermal crystallization process of AAH and can accelerate the polymerization process of ECH-AAH. Moreover, the low MW of ECH and AAH compared to the hybrid GO-ECH-AAH nanocomposite, POM photographs of the high MW of GO-ECH-AAH shows more black flakes, the result should contribute to the inferior disperse ability of hybrid ECH-AAH within the GO phases.

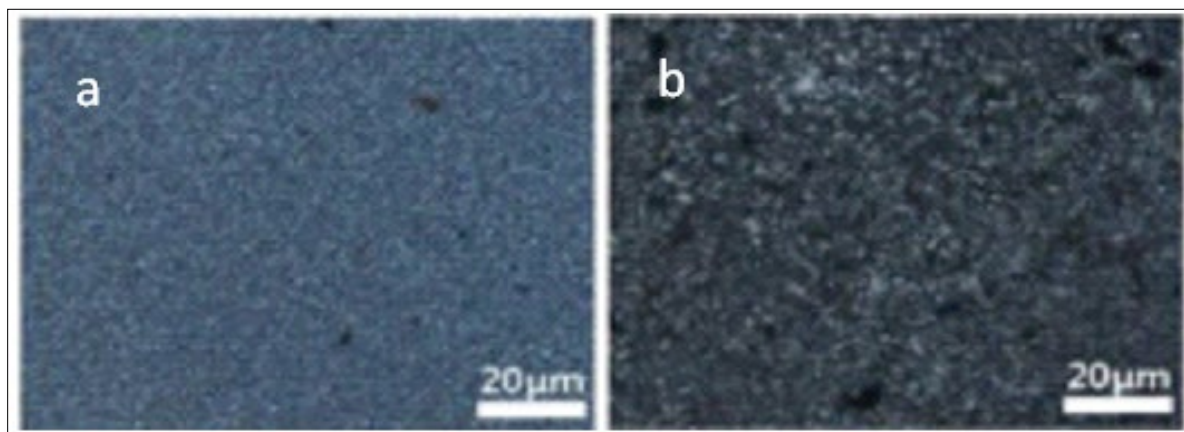


Figure 5. Polarizing microscope (POM) images of (a) Pure poly-allyl amine hydrochloride (AAL) and (b) GO-ECH-AAH nanocomposite.

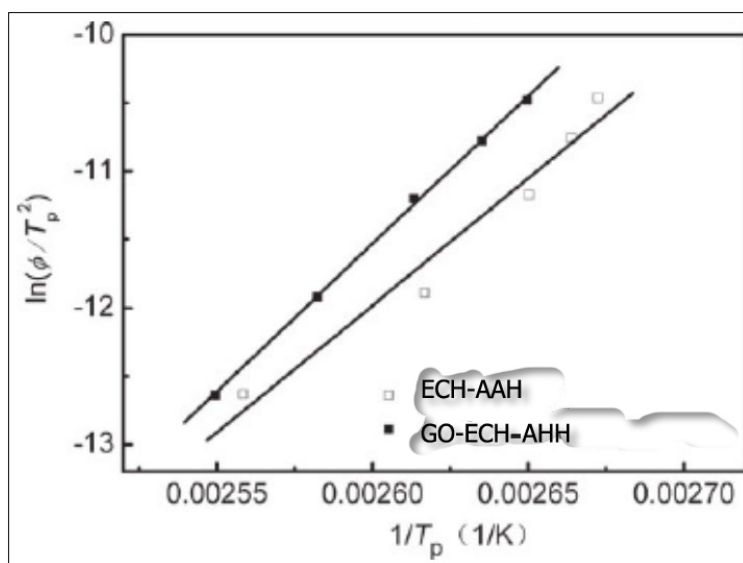


Figure 6. Polarized optical microscopy (POM) images of (a) pure poly-allyl amine hydrochloride (AAL) and (b) GO-ECH-AAH at 65°C during non-isothermal crystallization from their melts at a cooling rate of 2°C/min.

XPS scanning spectrum

The presence of functional groups GO surface was confirmed by XPS analysis. As expected, oxygen and nitrogen were found in GO exhibits high oxygen content

from its oxidation. Also, exploratory scans have indicated the residual presence of sulfur from growth and functionalization processes in. In our studies, ECH-AAH was grafted onto the GO sheets by *in situ* ring opening polymerization of GO. The grafted GO-ECH-AAH

dissolved well in dichloromethane, chloroform, DMF, THF, toluene and ethylene acetate. The homogeneous dispersion of GO in the polymer matrix improved the mechanical properties of ECH-AAH. The aggregation and stacking of GO nanosheets were also supported by tethering GO sheets on the ECH-AAH chains. We verified this morphology the presence of well-dispersed layers, indicating that after chemical reduction the material has not organized its crystal structure. Consequently, an increase in surface area of these nanoparticles in the nanocomposite, allowing the modification of the polymeric matrix structure, and this may result in increased elastic modulus and hardness of the sample; possibly change in the degree of crystalline because smaller nanoparticles can act as nucleation sites.

The presence of functional groups on GO surface before and after functionalization treatments of GO with ECH-AAH to form GO-ECH-AAH nanocomposite polymer surface was confirmed by XPS analysis (**Table 1**). As expected, oxygen and nitrogen were found in GO and GO-ECH-AAH exhibits high oxygen content from its oxidation. Also, exploratory scans have indicated the residual presence of sulphur and from growth and functionalization processes in GO-ECH-AAH, as shown in **Figure 7**, as well as residual presence of sulphur in GO from its synthesis process as shown in **Figure 8**.

Table 1. Atomic concentration from high resolution XPS analysis of GO-ECH-AAH and GO ($\pm 5\%$ precision).

Elements	Atomic concentration (%)	
	GO-ECH-AAH	GO
Carbon (C1s)	94	63.8
Oxygen (O1s)	25.1	35.1
Nitrogen (N1s)	1.9	1.1

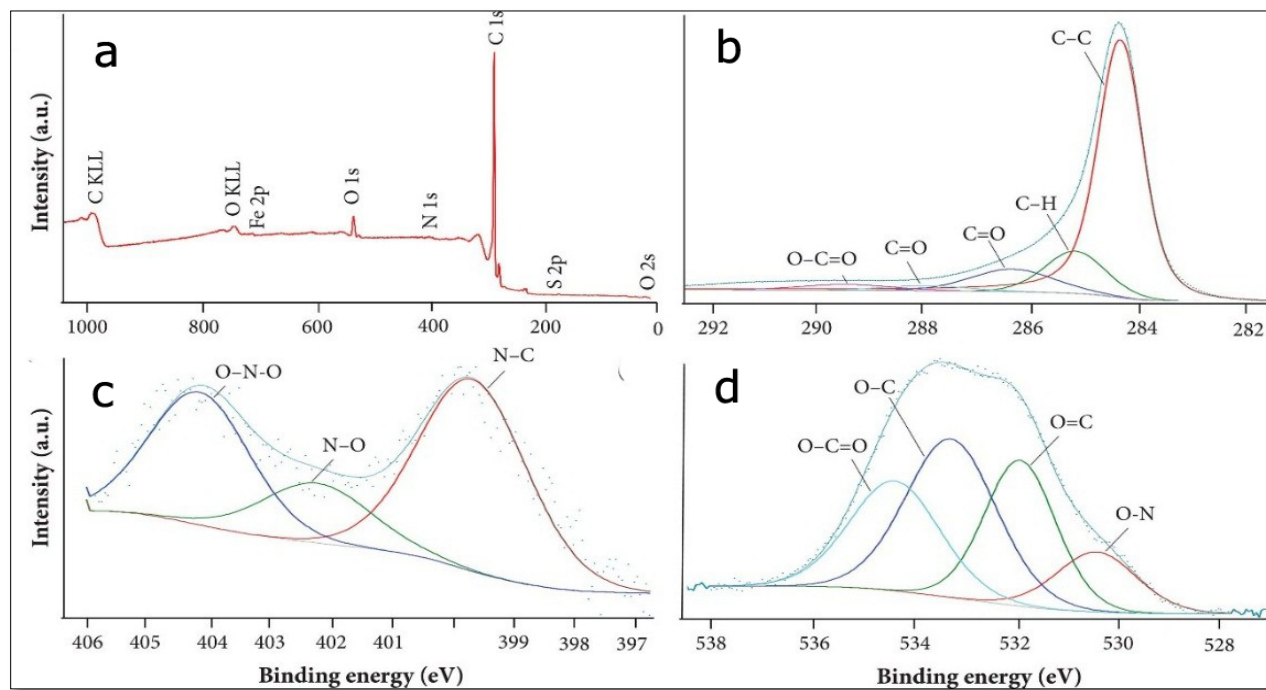


Figure 7. GO-ECH-AAH nanocomposite polymer XPS scanning spectrum (a) and XPS high-resolution survey scans of: (b) C1s spectrum; (c) N1s spectrum; (d) O1s spectrum.

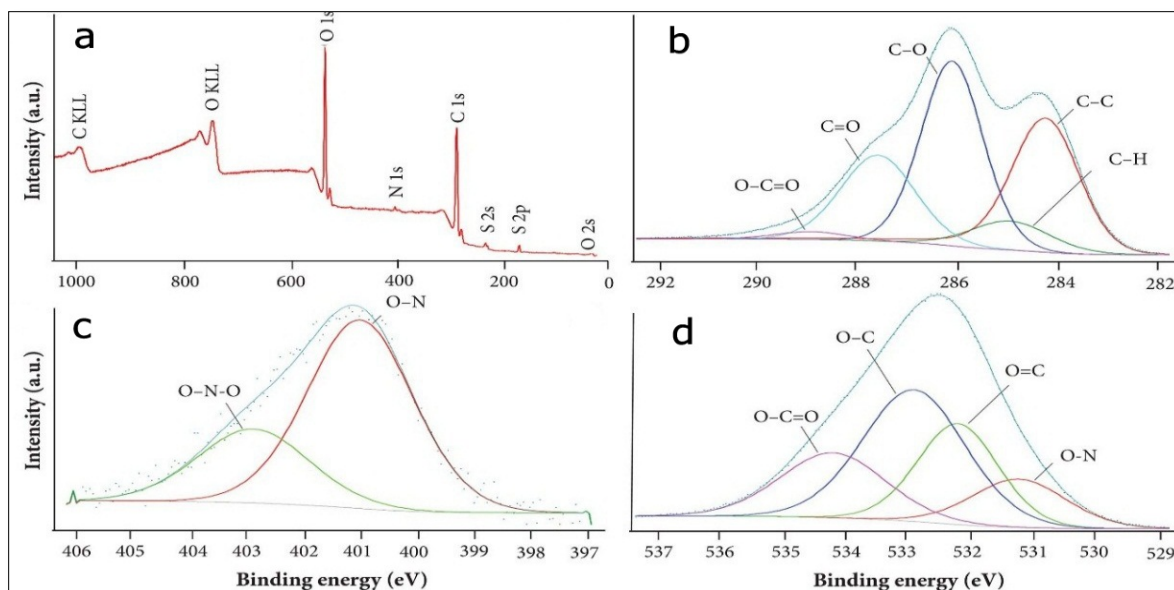


Figure 8. GO XPS scanning spectrum (a) and XPS high-resolution survey scans of: (b) C1s spectrum; (c) N1s spectrum; (d) O1s spectrum.

Raman spectrum of GO and GO-ECH-AAH nanocomposite

Figure 9 shows the Raman spectrum of GO and GO-ECH-AAH nanocomposite at different temperature as grown have clear bands at 1344.1 cm^{-1} (D band) and 1575.99 cm^{-1} (G band) and a shoulder near 1586.1 cm^{-1} (D' band). After functionalization, GO-ECH-AAH nanocomposite presents two bands at 1353.7 cm^{-1} (D band) and 1590.9 cm^{-1} (G band) and a shoulder near 1614.64 cm^{-1} (D' band). These bands are characteristic of multi-walled GO. The higher intensity of the G band for CNTs as grown indicates a higher degree of

graphitisation/crystallinity, while D band is typically attributed to disordered structures (defective GO and non-crystalline carbon). The change in intensities of D and G band could be observed on the Raman spectrum of GO-ECH-AAH due to the acid and amino functionalization process. It is known that during oxidizing treatments of graphitic structures two concurring phenomena take place: the removal of amorphous carbon from the GO and the formation of oxygenated functional groups, changing the atomic structure from C-C sp^2 to C-C sp^3 . Due to this change, a displacement in the position of G band and a higher intensity of D band can be observed.

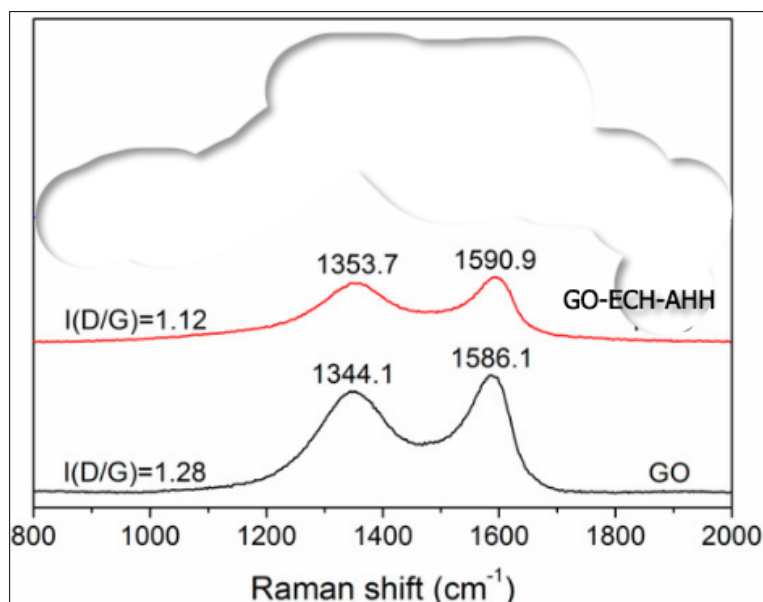


Figure 9. Raman spectrum of: (a) GO, and (b) GO-ECH-AAH nanocomposite at different temperature.

UV-Vis absorption spectra

Figure 10 shows the UV-vis absorption spectra of the GO, the ECH-AAH and GO-ECH-AAH nanocomposite. GO presented by characteristic peak at 229 nm corresponding to $\pi-\pi^*$ transitions of aromatic C-C bonds. Red shift peak of the graphene presented at 260 nm because the electronic conjugation in the graphene. While for GO-ECH-AAH nanocomposite, the absorption peak was shifted to 280 nm,

suggesting that the covalent attachment of ECH-AAH on to GO surface. As shown in **Figure 10**, there is a recorded blue shift in absorption maxima for the pure Allyl Amine Hydrochloride (AAH) polymer complex with ECH-AAH when complexed with GO sheet. I guess it is caused by complexation and an electron transfer between GO and ECH-AAH. So the active sites of GO sheet were blocked by complex formation with ECH-AAH polymer NPs.

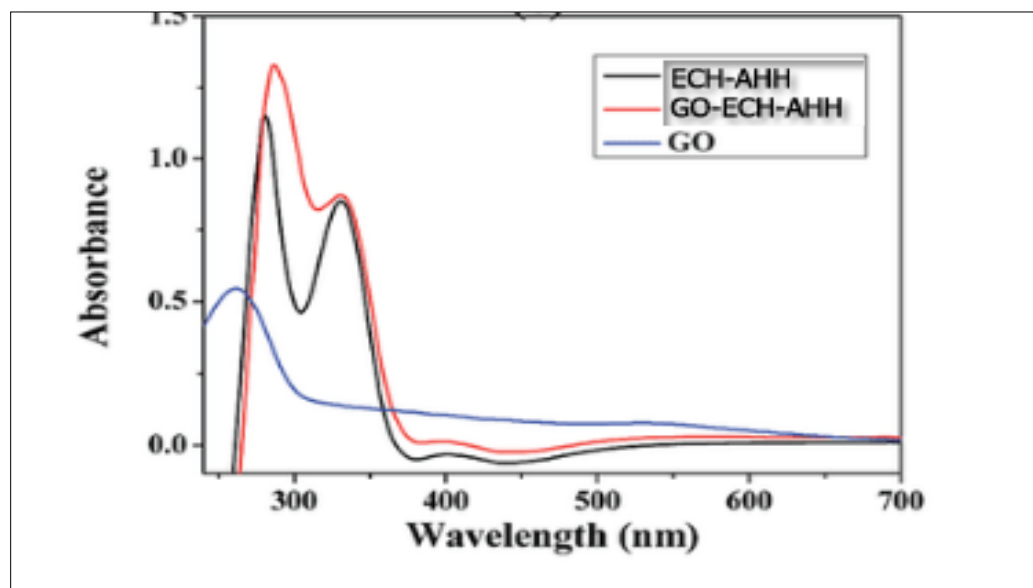


Figure 10. UV-vis absorption spectra of GO, the ECH-AAH and GO-ECH-AAH nanocomposite.

Crystallinity of carbonaceous materials can be evaluated by the ratio between D and G band intensity (ID/IG), as well as

full width at half height (FWHM) of G band. **Table 2** shows the information from the deconvoluted spectra.

Table 2. Deconvolution data of Raman spectrum of GO and GO-ECH-AAH nanocomposite.

Parameters	GO	GO-ECH-AAH nanocomposite
ID/IG	0.45	1.25
D band (cm^{-1})	1328.5	1348.33
G band (cm^{-1})	1575.99	1581.96
FWHM (D)	26.78	23.08
FWHM (G)	17.54	21.96

ID/IG ratio changed from 0.45 for GO as grown to 1.25 for functionalized GO-ECH-DETA nanocomposite. The increase in ID/IG ratio suggests that formation of oxygenated functional groups was more intense than removal of amorphous carbon.

FT-IR analysis

The FTIR spectra were obtained between 4000 and 350 cm^{-1} . **Figure 11** shows the FT-IR spectrum of (a) GO and (b) GO-ECH-AAH nanocomposite. The technique was used to investigate the structure and functional groups of both

samples. **Figure 11b** shows FT-IR spectrum of GO, shows adsorption bands at 1723 cm^{-1} , due to the C=O stretch of COOH group; at 1621 cm^{-1} , for stretch of C=C groups; at 1220 cm^{-1} , for C=C skeleton vibration; and at 1043 cm^{-1} for alkoxy C-O groups. Although graphite had been oxidized into GO, C=C groups led to the conclusion that the main structure of graphite layer was retained. The presence of oxygen-containing functional groups confirmed that the graphite was greatly oxidized into GO and was in agreement with the literature. FT-IR spectrum for GO-ECH-AAH nanocomposite sample (**Figures 7c and 7d**) shows the

following bands and peaks of interest: at 1670 cm^{-1} , corresponding to the amide carbonyl ($\text{C}=\text{O}$) stretching; at 3728 cm^{-1} , due to $-\text{NH}$ stretching; at 1587 cm^{-1} , because of N-H in-plane bending; and at 1220 and 1047 cm^{-1} , assigned to C-N stretching. These bands prove the presence of amide groups and lead to the conclusion that carboxylic groups on GOs surface were modified by amine from AAH. The band in the range of 3445.15 cm^{-1} showed a relatively broad bandwidth that is probably related to the axial deformation of the O-H bond due to the reaction with ECH. The other bands in the range of 3026.35 cm^{-1} C-H show that the nanocomposite formation process between the ECH, AAH and GO on the surface was successful. However, the characteristics and similar bands have the predominance of GO. The oxygen atoms tend to combine with carbon atoms

thereby forming an array of functionality, among which can be mentioned: ketones, esters, carboxylic acids and others. The three faint bands were observed in the region between 1651 and $1452\text{--}1366\text{ cm}^{-1}$ due to bending vibrations and axial deformation of the $\text{C}=\text{C}$ bonds is low because of GO with respect to the nanocomposite polymer matrix. The two most intense peaks has its stretching vibration ascribed to $\text{C}=\text{O}$ appeared in the range $1493\text{--}1601\text{ cm}^{-1}$ are due to the formation of hydroxyl and carboxyl groups, resulting from the chemical reaction. The band located at 748 cm^{-1} is related to the C -axial deformation the primary alcohols and other of band and 540 cm^{-1} are due to the angular deflection of C-H with H out of plane. Therefore, GOs were indeed functionalized through ECH and AAH treatments.

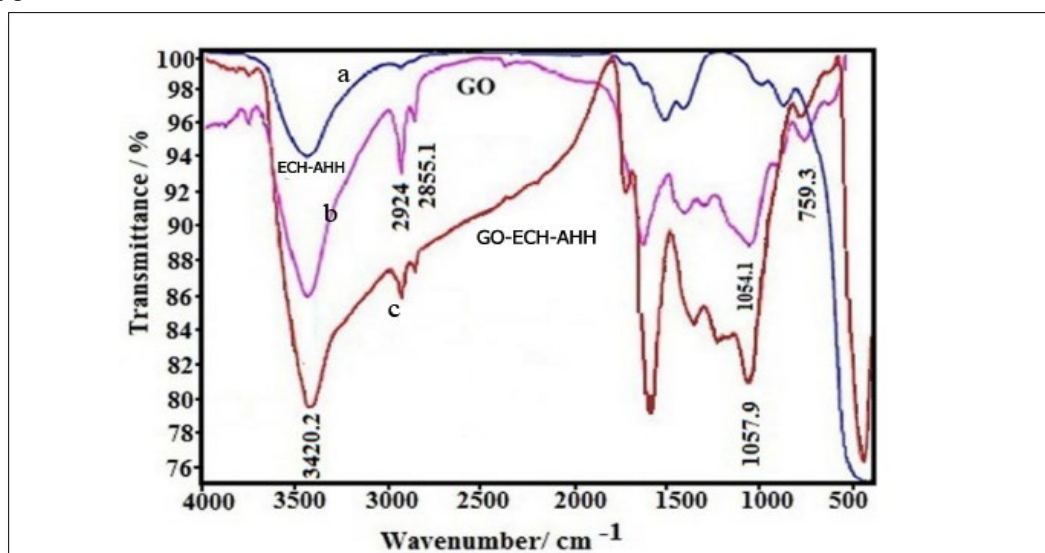


Figure 11. FT-IR spectrum of (a) GO, (b) ECH-AAH; and (c) GO-ECH-AAH nanocomposite.

LPS removal by GO-ECH-AAH nanocomposite

Several experiments were done to evaluate GO-ECH-AAH nanocomposite for their ability to eliminate LPS from aqueous solution and protein preparations. First, 50 ml protein solution (10 ppm) containing 200.0 endotoxin units per milliliter (EU mL^{-1}) was subjected to dialysis for 24 h against distilled water using a Slide-A-Lyzer 10 kDa cassette (Life Technologies, cat. # 66380). Second, a volume of 1 ml of 0.1% TX-114 in buffer and 1 ml of 1.0 mg/ml of BLG were applied to a HiTrap Desalting column (GE Healthcare Life Sciences, cat. # 29-0486-84). Third, samples were centrifuged at $21,000\text{ g}$ and 25°C for 10 and 20 min and at 37°C for 10 and 20 min. Fourth, samples were centrifuged using $0.22\text{ }\mu\text{m}$ (cat. # 8160) and $0.45\text{ }\mu\text{m}$ (cat. # 8162) spin X filter columns (Costar) at $10,000\text{ g}$ at 37°C for 4 min. Last, 50 mg of GO-ECH-AAH nanocomposite were tested for their efficacy to remove LPS from both pure aqueous solution and the protein solution, at 25°C . The LPS concentration was measured using the commercially available Endozyme Recombinant Factor C assay (Hyglos,

cat. # 609050) according to the protocol of the manufacturer. The measurements were conducted in triplicate using a Tecan Infinite 200Pro plate reader. The positive control (MQ) was spiked with a concentration of 0.45 EU/l of LPS, and results for all tests were considered valid when the value of recovered LPS concentration was between 50 and 200% of this value. LPS units were converted from EU (Endotoxin Unit) to concentrations in pg/mL by assuming that 1 EU corresponds to 100 pg of standard endotoxin EC-5. Thus, this is a simple and cheap procedure, has proven to remove endotoxins without affecting any significant losses in protein yields and biological activities. The initial concentration of LPS in both pure aqueous solution and the protein solution was estimated to be 200 EU/ml. GO-ECH-AAH nanocomposite LPS extraction reduced the LPS concentration in biological protein sample to 62.85 EU/ml and aqueous solution to 61.65, demonstrating in purification efficiencies of 68.58% and 69.18%, respectively (**Tables 3 and 4**). Repeated (one, two or three times) LPS-extraction did not lower LPS levels further significantly.

Table 3. The compared adsorption capacity of LPS from aqueous biological sample with GO and GO-ECH-AAH nanocomposite for nanocomposites. Endotoxin concentration of 200.0 endotoxin units per milliliter (EU mL⁻¹) and incubation time, 60 min.

pH	Results ^a ± S.D.	Results ^a ± S.D.
	Adsorption capacity of LPS using GO-ECH-AAH nanocomposite, EU mg ⁻¹ , from aqueous solution	Adsorption capacity of LPS using GO-ECH-AAH nanocomposite, EU mg ⁻¹ , from protein solution
4.0	132.40	130.34
5.0	135.34	131.78
6.0	138.35	137.15
7.0	137.23	136.05

a: Mean of three determinations

Table 4. The compared adsorption capacity of LPS from aqueous biological sample with GO and GO-ECH-AAH nanocomposite for nanocomposites. Endotoxin concentration of 200.0 endotoxin units per milliliter (EU mL⁻¹) and pH 6.0.

Incubation Time, min	Results ^a ± S.D.	Results ^a ± S.D.
	Adsorption capacity of LPS using GO-ECH-AAH nanocomposite, EU mg ⁻¹ , from aqueous solution	Adsorption capacity of LPS using sing GO-ECH-AAH nanocomposite, EU mg ⁻¹ , from protein solution
30	122.10	120.46
60	129.45	128.23
90	138.63	134.32
120	135.28	131.31

CONCLUSION

In summary, we developed multi-functional GO-based nanocomposites, GO-ECH-AAH, by the reaction of graphite oxide (GO) with epichlorohydrin (ECH) as a coupling agent and allylamine hydrochloride (AAH) a ligand. The instrumental analysis of the nanocomposite prove that there is chemical interaction between ECH, AAH and GO on the surface of GO. For example, the presence of amide groups and lead to the conclusion that carboxylic groups on GOs surface were modified by amine from AAH. The band in the range of 3445.27 cm⁻¹ showed a relatively broad bandwidth that is probably related to the axial deformation of the O-H bond due to the reaction with ECH. The other bands in the range of 3026.55 cm⁻¹ C-H show that the nanocomposite formation process between the ECH, AAH and GO on the surface was successful. The polymeric chains on the surface of GO endowed GO-ECH-AAH nanocomposite with excellent chain stability. Additionally, from the SEM images we observed the interface between the GO and the epoxy composite ECH-AAH suggests chemical interaction. As can be seen from this research, the GO-ECH-AAH

nanocomposite can be used as a potential candidate for removal of PLS from aqueous solution.

CONFLICTS OF INTEREST

The authors declare no conflict of interest regarding the publication of this paper.

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